

Crystallization kinetics of nanoconfined GeTe slabs in GeTe/TiTe₂-like superlattices for phase change memories

Supplementary Information

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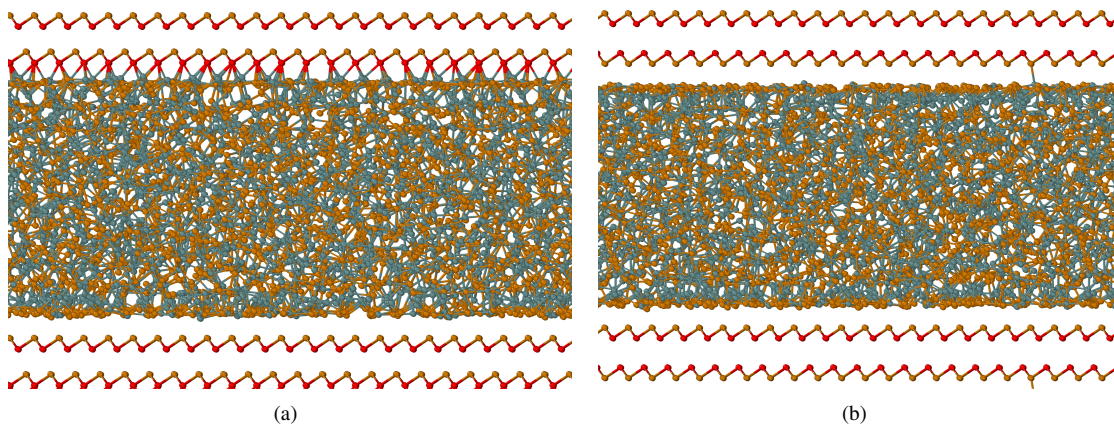


Figure S1. Two models of the amorphous phase of a slab initially made of nine bilayers of α -GeTe encapsulated by capping layers. The capping layers are made of a frozen bilayer of crystalline GeTe at the lattice constant of TiTe₂ aiming at mimicking the confining slabs of TiTe₂ in GeTe/TiTe₂ superlattices. (a) In the first model, the TiTe₂-like bilayer was oriented in such a way that Ge on one side and Te on the other side face the crystalline GeTe block with an interplanar distance between the capping layer and the outermost plane of the GeTe block of 3.03 Å. (b) In the second model, the TiTe₂-like bilayer was oriented in such a way as to expose the Te layer to the crystalline GeTe block on both sides. The distance between the capping layer and outermost layer of GeTe slab is 3.40 Å for the Te layer and 2.74 Å for the Ge layer. The color code is the same of Fig. 1 in the article.

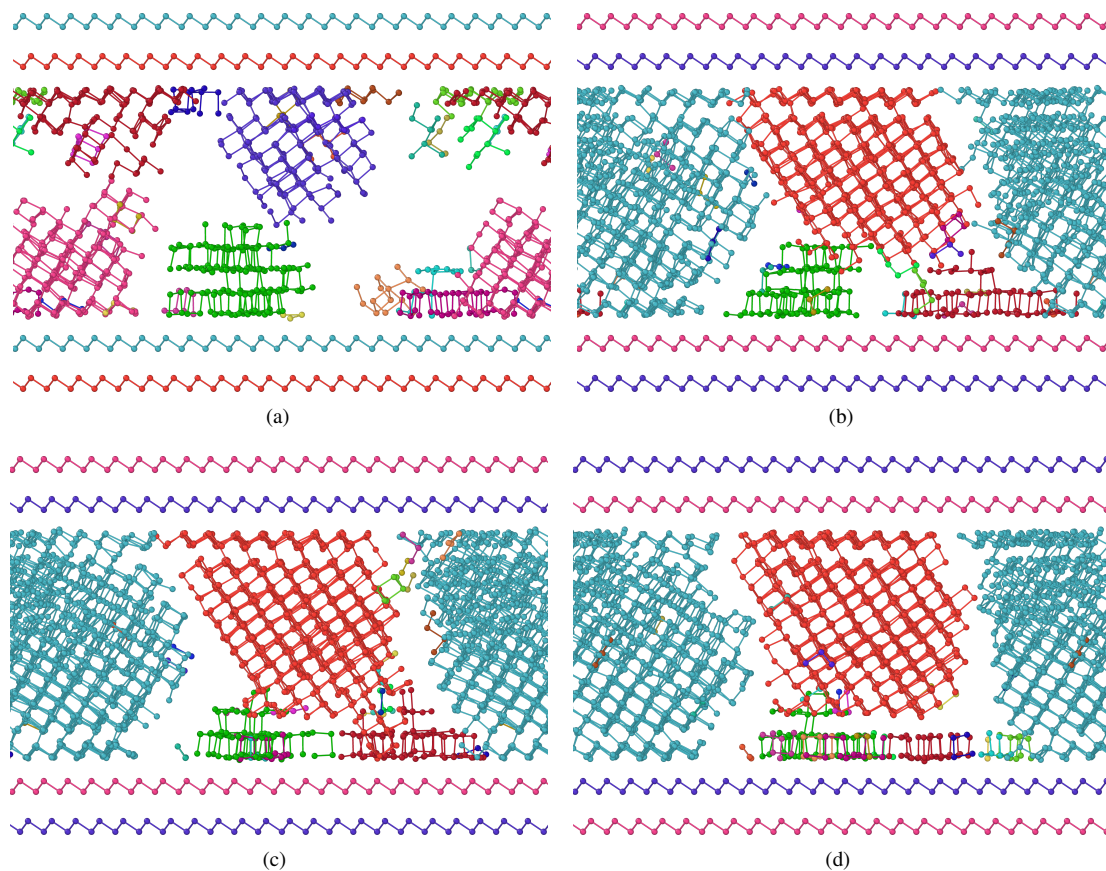


Figure S2. Snapshots of crystalline atoms of the GeTe/TiTe₂-like superlattice identified by the bond parameter Q_4^{dot} at 650 K. Different crystalline nuclei are shown with different colors. The snapshots are shown for (a) 0.25 ns (b) 0.5 ns (c) 0.75 ns and (d) 1 ns.

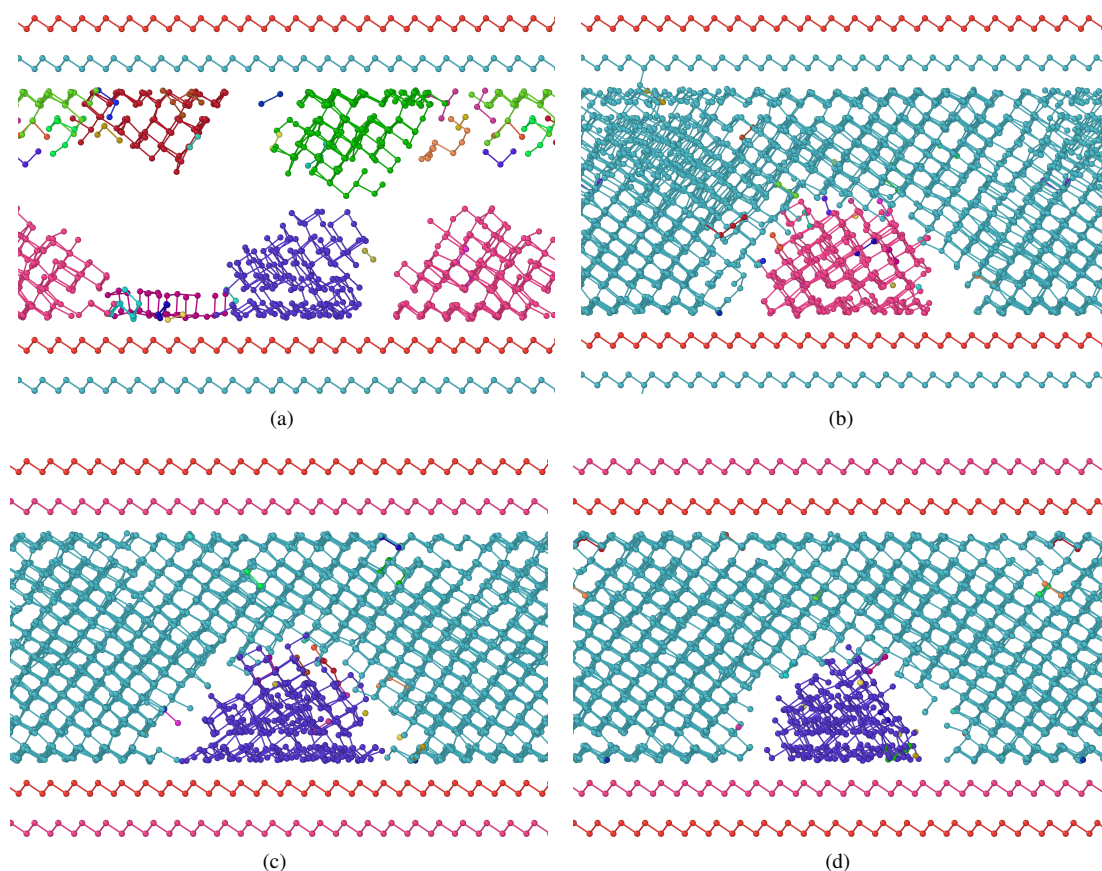


Figure S3. Snapshots of crystalline atoms of the GeTe/TiTe₂-like superlattice identified by the bond parameter Q_4^{dot} at 700 K. Different crystalline nuclei are shown with different colors. The snapshots are shown for (a) 0.25 ns (b) 0.5 ns (c) 0.75 ns and (d) 1 ns.

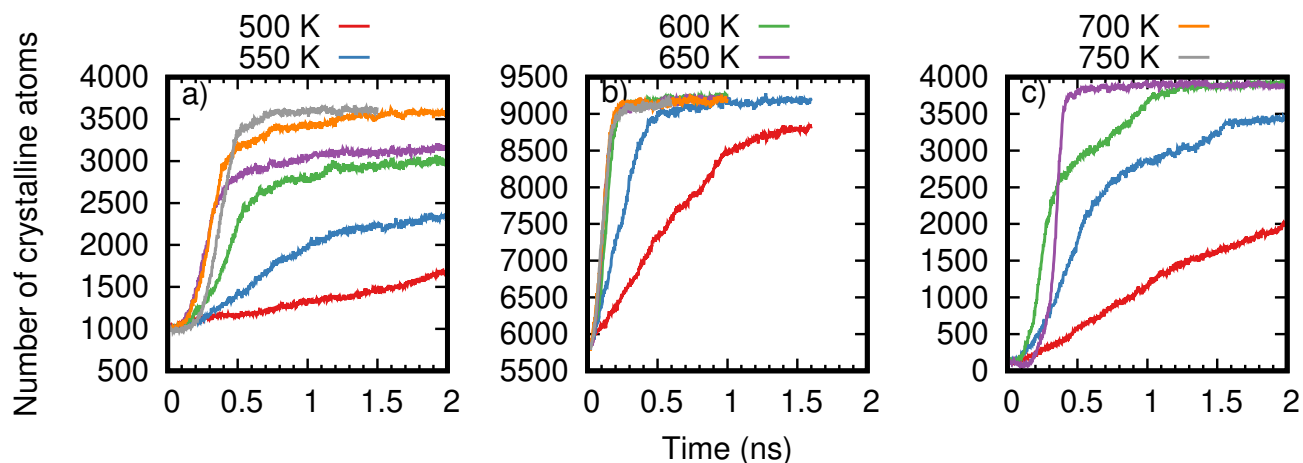


Figure S4. Number of crystalline atoms identified with the order parameter Q_4^{dot} (see article) as a function of time at different temperatures for (a) GeTe/TiTe₂-like superlattice (b) bulk a-GeTe (model Bulk28, see article) with crystal growth from the crystal/liquid interface and (c) bulk a-GeTe with homogeneous crystal nucleation and growth.

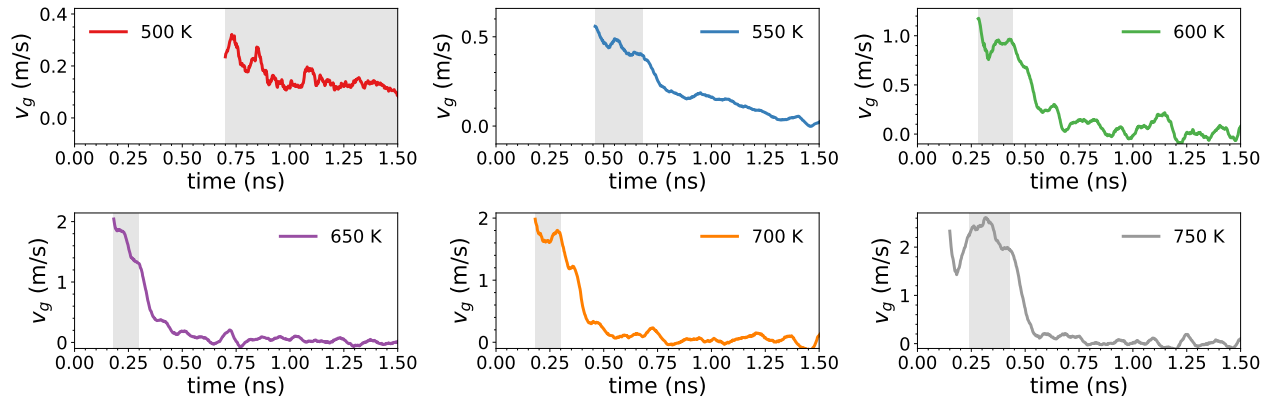


Figure S5. Instantaneous crystal growth velocity as a function of time for the model of the GeTe/TiTe₂-like superlattice at different temperatures. The region highlighted in gray corresponds to the time interval over which we estimated the average crystal growth velocities reported in Table 2 in the article. We start computing the instantaneous velocity when all the crystalline nuclei become overcritical.

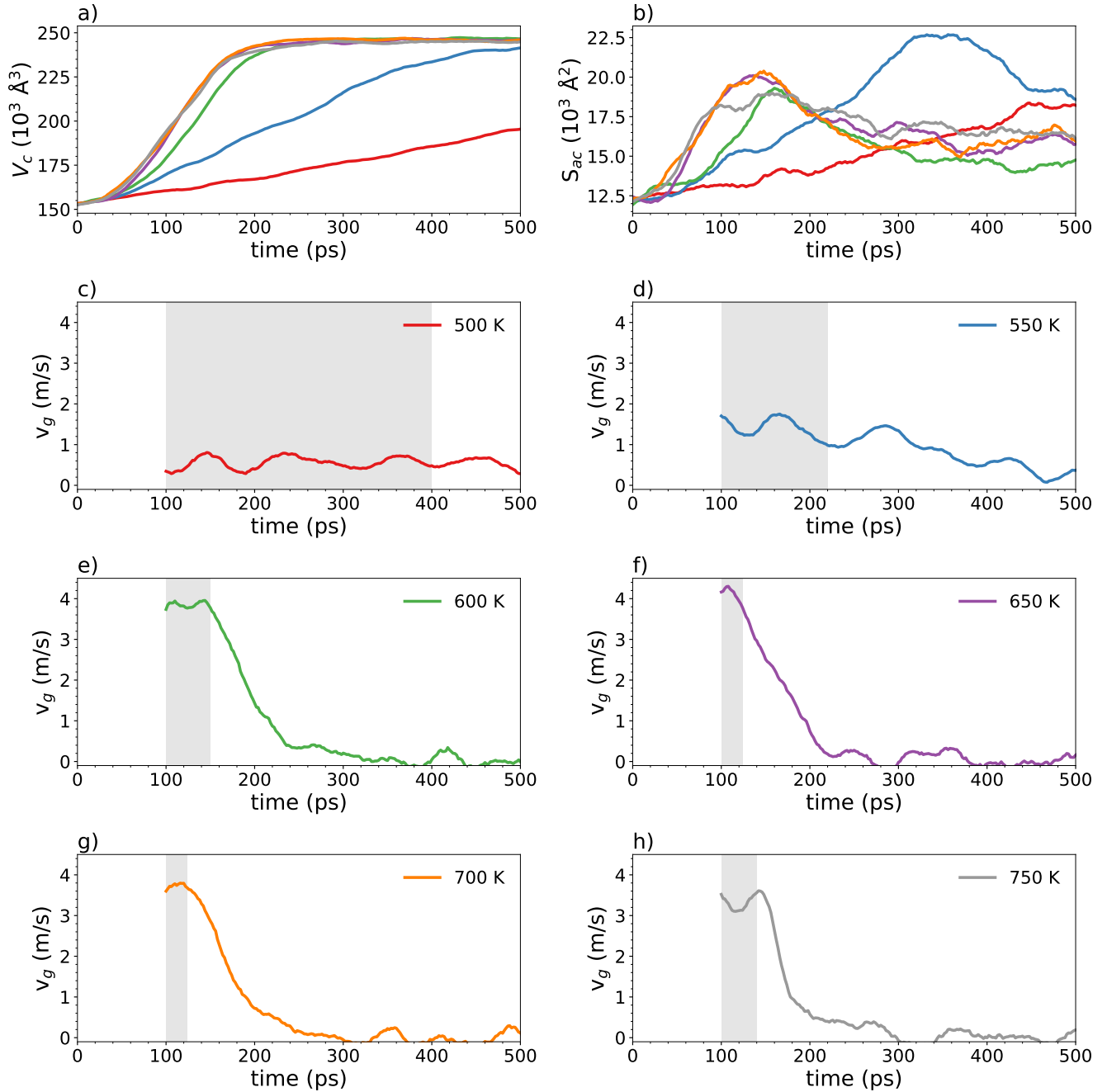


Figure S6. (a) The evolution in the time of the crystalline volume (V_c) and of (b) the area of the amorphous-crystal interface (S_{ac}) at different temperatures in bulk a-GeTe for crystal growth from the amorphous/crystal interface (model Bulk28, see article). (c)-(h) instantaneous crystal growth velocities computed as $v_g = dV_c/dtS_{ac}^{-1}$ as described in Ref. 1. The region highlighted in gray in panels (c)-(h) corresponds to the time interval over which we estimated the average crystal growth velocities reported in Table 2 in the article. We started computing v_g at a later time after proper thermalization of the system.

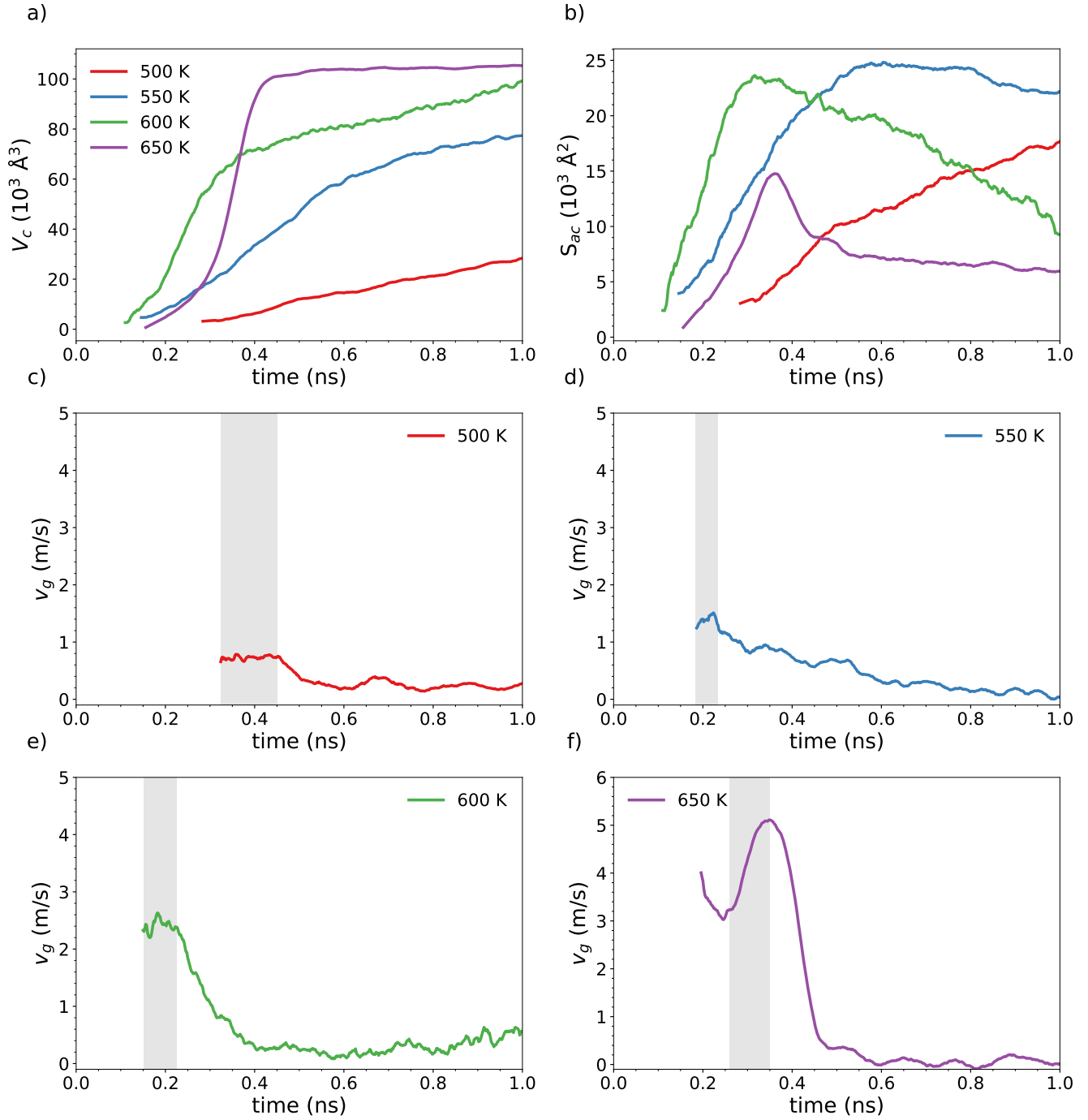


Figure S7. (a) The evolution in time of the crystalline volume (V_c) and of (b) the area of the amorphous-crystal interface (S_{ac}) at different temperatures in the 4096-atom model of bulk a-GeTe for homogeneous crystal nucleation and growth (see article). (c)-(h) instantaneous crystal growth velocities computed as $v_g = dV_c/dtS_{ac}^{-1}$ as described in Ref. 1. The region highlighted in gray in panels c)-h) corresponds to the time interval over which we estimated the average crystal growth velocities reported in Table 2 ($Bulk_{homo}$) in the article. We start computing the instantaneous velocity when all the crystalline nuclei become overcritical.

Table S1. Two-dimensional diffusion coefficient D as a function of time of the confined GeTe slab in GeTe/TiTe₂-like superlattice extracted from NVE simulations at the average temperatures given in the first column. We computed D from the two dimensional mean square displacement in the plane perpendicular to the slab thickness in the superlattice as $\langle x^2 \rangle + \langle y^2 \rangle = 4Dt$. The diffusion coefficient in the slab is compared to those in the bulk at the experimental density of the amorphous phase from Ref. 2. The values for D in the bulk at the density of the slab which is close to that of crystalline α -GeTe are expected to be lower than those reported in the last column of the table which correspond to the experimental density of the amorphous phase.

Temperature (K)	D ($10^{-6}\text{cm}^2/\text{s}$)	
	GeTe/TiTe ₂ -like superlattice	Bulk from Ref. 2
512	0.6	1.5
561	1.2	2.6
614	2.8	4.5
652	4.9	6.2
709	8.2	9.4
750	12.9	12.3

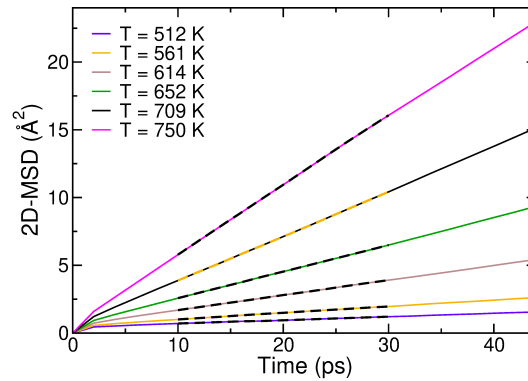


Figure S8. Two-dimensional mean square displacement (MSD) as a function of time from NVE simulations at the average temperatures given in the inset.

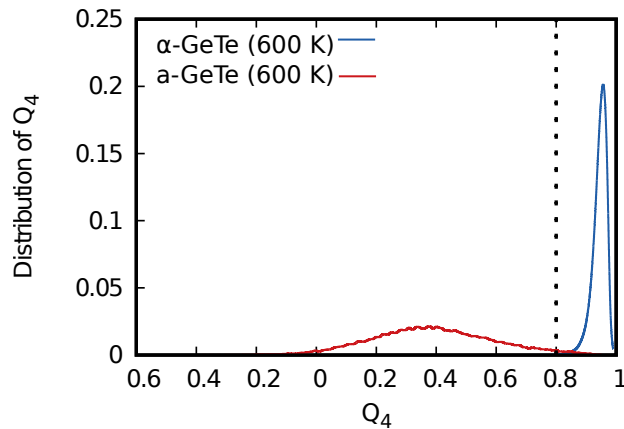


Figure S9. The normalized distribution of the Q_4^{dot} order parameter for amorphous GeTe overheated at 600 K (red curve) and for crystalline α -GeTe at 600 K (blue curve). The vertical dashed line sets the threshold to identify crystalline atoms.

References

1. Xu, Y. *et al.* Unraveling crystallization mechanisms and electronic structure of phase-change materials by large-scale ab initio simulations. *Adv. Mater.* **34**, 2109139, DOI: <https://doi.org/10.1002/adma.202109139> (2022).
2. Gabardi, S. *et al.* Atomistic simulations of the crystallization and aging of GeTe nanowires. *J. Phys. Chem C* **121**, 23827–23838, DOI: [10.1021/acs.jpcc.7b09862](https://doi.org/10.1021/acs.jpcc.7b09862) (2017).