

Laboratory simulation of space weathering on silicate surfaces in water environment

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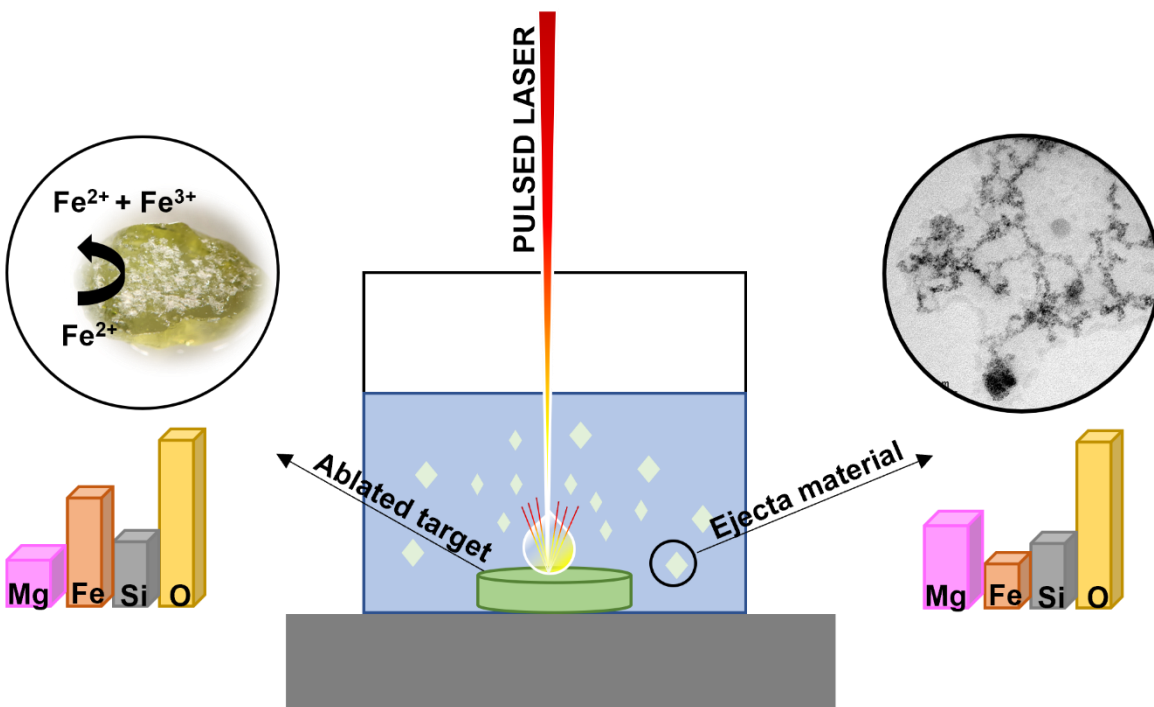
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ABSTRACT

Silicate nanoparticles occur in various astrophysical environments where they experience substantial processing due to events such as grain-grain collisions and irradiation. However, the structure and chemical evolution together with the origin of these grains are still poorly understood and intensively debated. For this purpose, we performed liquid-phase nanosecond pulsed laser ablation on olivine single crystals to (i) simulate space weathering in a water environment (e.g., hydrous or volatile-rich bodies) and (ii) study the chemical and structural evolution of both the target surface and the ablated material. In particular, optical spectroscopy analyses have been performed on the ablated material and correlated with high-resolution transmission electron microscopy and diffraction; whereas, compositional variations of the ablated target surface were determined by X-ray photoelectron spectroscopy. Our results show that the target material is

19 enriched in Fe and depleted in Mg after the ablation process, with the water environment triggering
20 the oxidation of Fe^{2+} into Fe^{3+} in a region confined at the solid-liquid interface and thus promoting
21 the formation of magnetite on the sample surface. On the other hand, in the ablated material we
22 find olivine crystalline fragments with shock features together with Mg-rich crystalline
23 nanoparticles. Notably, no metallic iron nanoparticles have been detected in the ablated material.
24 Our simulation of space weathering in water environment revealed structural and chemical
25 changes which are expected to give rise to distinctive features in the reflectance spectra when
26 compared to those from airless bodies of the inner Solar System.

27 GRAPHICAL ABSTRACT



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30

31 INTRODUCTION

32 Dust grains occur in various astrophysical environments (i.e., from comets to the farthest
33 known galaxies) being the major responsible of a large variety of processes in the interstellar
34 medium ^{1,2}. Indeed, thanks to their high surface-to-volume ratio, they act as catalytic substrates for
35 the formation of common and complex molecules and their structure (i.e., crystalline or
36 amorphous), size and shape, play a fundamental role in the surface reactivity ^{3,4}. In particular,
37 according to ref ⁵, interstellar grains are divided into three main size domains in accordance with
38 the ranges 1000-100 nm, 100-5 nm, and 5-0.3 nm. In the context of materials science and
39 engineering, particles with a size included in the last two ranges are referred to as nanoparticles, a
40 term used to indicate a substantial distinction of the material properties with respect to those of the
41 bulk phase. Furthermore, particles with a size included in the last range are also usually referred
42 to as Very Small Grains (VSGs), in accordance with the terminology employed in planetary
43 sciences. VSGs are those with the highest relative abundance among interstellar grains.

44 Among dust grains, the silicates are the most common species ^{6,7}. The existence of
45 interstellar silicate grains was first proposed by Kamijo in 1963 ⁸ and a few years later they were
46 detected in several cosmos regions. The broad infrared (IR) feature detected around 10 μm
47 corresponding to the Si-O stretching mode suggests that most silicate grains in the interstellar
48 medium are amorphous. However, the new advanced spectrometers coupled with the development
49 of the Infrared Space Observatory (ISO) revealed the presence of crystalline silicates around
50 comets, dust disks and in interplanetary dust particles (see ref ⁹ and references therein). These
51 grains experience a strong structural modification during their lifetime in the diffuse interstellar
52 medium due to events such as grain-grain collisions and irradiation. Grain amorphization is one of
53 the major effects, transforming crystalline silicate grains concentrated in star envelopes into

54 amorphous material populating the interstellar medium ¹⁰. However, after the amorphization
55 process, residual crystalline features could be still retained giving rise to a population of crystalline
56 silicate nanoparticles occurring in the interstellar regions ¹¹. Moreover, it cannot be ruled out also
57 the possibility that the crystalline size domains of the nanoparticles are below the detection limit
58 of the available spectroscopic techniques.

59 The main building blocks of the interstellar silicates are O, Si, Fe, Mg, Al and Ca; all
60 elements that are among the principal constituents of the Earth's surface ¹², thus leading to the
61 name "astronomical silicates". Moreover, the comparison of acquired IR spectra with laboratory
62 ones provided the necessary information to confirm that astronomical silicates have mainly olivine
63 and pyroxene composition ^{7,11,13,14}.

64 The persistent space weathering processes these astronomical silicate grains experience in
65 the outer space (e.g., from irradiation, ion bombardment to impact shock events) might cause also
66 irreversible chemical changes ¹⁵. These changes are still poorly understood and characterized given
67 the difficulties in reproducing the pertinent atmospheres and environments and the limited
68 instrumental resolution of remote sensing data. However, understanding the effects of these
69 processes is of fundamental importance (i) for the correct interpretation of remote sensing data and
70 (ii) to increase the knowledge on nanoparticles vs bulk material properties ⁷.

71 For this purpose, space weathering processes have always been the subject of experimental
72 and theoretical studies ^{1,15-19} aimed at simulating the physical and optical properties of interstellar
73 dust grains and the surface alteration of planetary bodies ^{4,10,11,20-23}. In particular, irradiation with
74 high energy ions (e.g., H⁺, He⁺), nano- and femtosecond pulsed laser ablation have been carried
75 out in vacuum on selected silicate samples (e.g., powder, single crystals, pellets) to reproduce the
76 effects of solar wind radiation and shock events on airless bodies ^{10,16,17,22,24-26}.

77 Furthermore, hydrous or volatile-rich bodies with distinct amounts of H and OH such as
78 Vesta, Ceres, are also subjected to space weathering processes ^{15,27-30}. However, these bodies,
79 despite being airless, do not exhibit a measurable content of metallic iron particles on their surface.
80 This evidence has been attributed to a lesser flux of solar wind and meteoroid bombardment with
81 respect to Moon or Mercury ^{15,29}. Moreover, dust/ice agglomerates are known to populate dense
82 molecular clouds, comets and planet-forming disks (e.g., evidence from samples returned by the
83 Stardust mission ³¹ and data obtained from the Rosetta mission ³²). For examples, the recent results
84 by Potapov et al. ^{33,34} showed that silicate grains can entrap water ice molecules at 8 K giving rise
85 to stable silicate agglomerates mixed with ice up to 200 K when the water molecules start to desorb.

86 Even though space weathering processes are receiving more and more attention, the role
87 of volatile species in space weathering together with the structural-chemical evolution and the
88 origin of these cosmic silicates are still poorly understood and intensively debated ^{7,30,35}.

89 Therefore, the aim of this study is the experimental simulation of space weathering
90 processes by liquid-phase nanosecond pulsed laser ablation (LP-PLA) on olivine single crystals to
91 shed light on the structure and chemical evolution of both the impacted silicate target and the
92 ejected/ablated material in a water environment.

93 Laser ablation is usually performed in vacuum or inert atmosphere, but an alternative
94 configuration was proposed by Patil et al. ³⁶ where the target is immersed in a liquid medium, and
95 the high intensity laser beam is focused through the liquid onto the surface of the target. This
96 procedure is also widely used to synthesize nanoparticles in a clean environment with respect to
97 chemical synthesis methods, thus providing several advantages in both laboratory and industrial
98 applications ^{37,38}. In LP-PLA, the interaction of the high intensity laser pulse with the target surface
99 produces an ablation plume of ejected material, in which the surface of the solid target and a small

100 amount of the surrounding liquid are vaporized to form bubbles until they collapse with substantial
101 increment of the cooling rates with respect to air or vacuum, giving rise to products in the form of
102 nanoparticles suspended within the liquid medium. This allows either for their simple retrieval by
103 filtration and evaporation of the liquid, or for their further laser processing³⁹. The tamping effect
104 of the liquid layer allows for the control of the ablation rate while incrementing local pressure,
105 with some advantages in terms of control of nanoparticle size distribution, shape and structure^{40,41}.
106 On the other side, the remnant target represents a useful product of the extreme conditions induced
107 during the plasma recoil effects (laser peening).

108 This work has been carried out by means of a multiple technique characterization in order
109 to investigate the chemical-physical properties of both the shocked target surface and the ablated
110 nanoparticles. X-ray Photoelectron Spectroscopy (XPS) analyses have been carried out to analyze
111 the surface chemical composition and speciation of the target material before and after laser
112 treatment. High-resolution transmission electron microscopy (HR-TEM) coupled with optical
113 spectroscopy and dynamic light scattering measurements have been performed on colloidal
114 solutions of the ablated nanoproducts. Analyses of their electrostatic properties and acid/base
115 reactivity have been also carried out for a deeper chemical characterization. Our results show that
116 a substantial chemical processing might be expected in both interstellar grains and silicate surfaces
117 during space weathering processes in water environments, having distinctive features with respect
118 to space weathering occurring in vacuum environment. This study provides also additional tools
119 for (i) the identification of grains of different origin by means of distinctive features in the
120 reflectance spectra (e.g., presence or absence of reddening phenomena) and (ii) the comprehension
121 of the cosmic evolution of dust particles with particular emphasis on the role of water.

122

123 METHODODOLOGY

124 The olivine crystals of this study (average size $5 \times 4 \times 3 \text{ mm}^3$ up to 1 cm for the longest
125 direction) were sampled along the Mosteiros beach (São Miguel island, Azores) from basaltic and
126 pyroclastic deposits of the Sete Cidades volcano. The Sete Cidades volcano lithologies spread from
127 basaltic to trachytic passing from the pre-caldera to the caldera-forming phase and the post caldera
128 (see ref ⁴²). The mafic rocks are classified as alkali-basalts and mainly contain phenocrysts of
129 olivine, clinopyroxene and plagioclase in a fine-grained matrix. Olivine phenocrysts of mm size
130 show euhedral shapes, a pale green color and their transparency is often reduced by the presence
131 of inclusion phases such as spinel, plagioclase and ilmenite ⁴².

132 A set of three Mosteiros olivine crystals has been selected for chemical analyses by
133 Scanning Electron Microscopy (SEM) Energy Dispersive X-ray Spectroscopy (EDS) to evaluate
134 the homogeneity of the bulk chemistry and the mineral chemistry of the inclusion phases. Whereas
135 another olivine crystal has been analyzed using Synchrotron Mössbauer spectroscopy to explore
136 the possible occurrence of Fe^{3+} in the target material. Pulsed laser ablation was performed on one
137 single crystal of olivine and high-resolution transmission electron microscopy followed by optical
138 absorption spectroscopy, dynamic light scattering and ζ -potential analyses were performed on the
139 resulting ablated material (see supporting info for more details about the dynamic light scattering
140 and ζ -potential measurements). XPS measurements were carried out in parallel on another olivine
141 crystal in order to evaluate the chemical composition variations before and after the ablation
142 process.

143 **Scanning Electron Microscopy**

144 The three selected crystals were embedded in epoxy and carbon coated for the SEM- EDS
145 analyses. The concentrations of major elements and, in some cases, NiO were measured by means

146 of a Zeiss Gemini 500 instrument operating at 20 keV and equipped with a Bruker QUANTAX
147 energy dispersive/wavelength dispersive detector.

148

149 **Synchrotron Mössbauer Spectroscopy**

150 Synchrotron Mössbauer spectroscopy was performed at the Nuclear Resonance Beamline
151 ID18⁴³, European Synchrotron Radiation Facility (Grenoble, France) using the setup reported by
152 Potapkin et al.⁴⁴. Room temperature Mössbauer spectra at the core and at the rim were collected
153 on one Mosteiros olivine ($0.5 \times 0.5 \times 0.3 \text{ mm}^3$). The experiments were carried out at ambient
154 conditions with Be lenses and at velocities of $\pm 5 \text{ mm s}^{-1}$ calibrated using a 25 μm thick natural α -
155 iron foil. The X-ray beam-size was $9 \times 4 \mu\text{m}^2$. The linewidth was controlled before and after each
156 sample measurement using $\text{K}_2\text{Mg}^{57}\text{Fe}(\text{CN})_6$, whose Mössbauer spectrum consists of a single line.
157 The data were then fitted with the software MossA using the full transmission integral and a
158 Lorentzian-squared source line shape⁴⁵.

159

160 **Liquid phase nanosecond pulsed laser ablation**

161 One olivine single crystal was selected for the LP-PLA experiment and it was placed at the
162 bottom of a polystyrene box filled with 4 ml of deionized water (type 1) to immerse it completely,
163 with the liquid level at ca. 3 mm above it. LP-PLA experiments were performed with a Nd:YAG
164 laser at 1064 nm focused via a singlet lens onto the surface of the target and operating at 5 ns and
165 10 mJ per pulse with a repetition rate of 10 Hz. The ablation experiment was performed by
166 manually moving the sample to perform a laser raster scan of the sample surface. In this procedure
167 multiple laser ablations on the same spot might occur. The ablation duration was of about 10 hours.

168

169 **X-ray photoelectron spectroscopy**

170 XPS analyses were carried out at the Physics Department of the Politecnico di Milano on
171 an olivine single crystal before and after the ablation. XPS measurements were performed in an
172 ultrahigh-vacuum system (the base pressure was in the low 10^{-10} Torr range) equipped with a 150
173 mm hemispherical analyzer from SPECS GmbH operating at a pass energy of 20 eV ⁴⁶.
174 Photoelectrons were excited by Al-K α radiation (1486.6 eV), yielding an overall full width at half-
175 maximum resolution of about 1 eV. The emitted electrons were collected at normal emission with
176 the sample kept at room temperature. Wide energy scans and detailed scans on selected core level
177 lines expected for the most abundant chemical species in the olivine structure (i.e., Si, Mg and Fe)
178 were acquired.

179

180 **Optical absorption spectroscopy**

181 Absorption measurements on the colloidal solution have been carried out in the range 200-
182 1100 nm (6.20 eV - 1.13 eV) and compared against reference colloidal solutions dispersed in
183 deionized water (i.e. mesoporous silica [SiO₂] nanoparticles, brucite [Mg(OH)₂] nanoparticles,
184 aluminum hydroxide [Al(OH)₃] nanoparticles, chrysotile [Mg₃Si₂O₅(OH)₄] nanotubes, and
185 synthetic forsterite [Mg₂SiO₄] nanoparticles). Moreover, additional absorption analyses have been
186 carried out as a function of the addition of known aliquots of sulfuric acid to study the suspensions
187 behavior as a function of pH.

188

189 **Electrodeposition and transmission electron microscopy**

190 High-resolution transmission electron microscopy (HR-TEM) and EDS analyses were
191 performed on the ablated nanoparticles deposited (i) directly from water suspension (2 μ l) and (ii)

192 via electro deposition of the dried water suspension re-dispersed in ethanol (applied voltage: 20 V,
193 electrode separation: 1 cm, duration: 20 minutes) onto C-coated Cu grids fixed by polypropylene
194 clamps on 1×1 cm² indium-tin-oxide (ITO)-coated glass plates. TEM analyses have been
195 performed with a JEOL JEM 2100Plus instrument at the Microscopy Platform of the University
196 of Milano-Bicocca. The instrument is equipped with a 9 MP Gatan Rio CMOS camera for image
197 acquisition and an Oxford EDS system for chemical analysis. The instrument was operating at 200
198 kV with a LaB₆ source. Image processing was carried out using the GATAN Digital Micrograph
199 software. In particular, direct measurement of the reflections from the selected area electron
200 diffraction (SAED) patterns, processing of the SAED patterns using the DiffTools⁴⁷ and the direct
201 measurement from the high-resolution bright field images coupled with the Fast Fourier Transform
202 (FFT) have been performed in order to determine the interlayer spacings of crystalline
203 nanoparticles and fragments.

204

205 RESULTS

206 Scanning electron microscopy analyses and Synchrotron Mössbauer spectroscopy

207 The olivine crystals from the “Mosteiros suite” are forsterite-rich [\sim For₈₇; (Mg_{1.74}, Fe_{0.26})
208 SiO₄] and contain plagioclase (NaAlSi₃O₈-CaAl₂Si₂O₈) and ilmenite (FeTiO₃) as inclusions
209 phases. The FeO content for all the samples is constant and around 13 wt% (see Table S1), while
210 Fe³⁺ was not detected by Synchrotron Mössbauer spectroscopy (Figure S1) and Ni concentration
211 is around 1300 ppm. These results are in agreement with chemical data reported by Beier et al.⁴²
212 on Azores olivines from the pre-caldera stage.

213

214 X-ray photoelectron spectroscopy

215 Wide XPS spectra of olivine before and after the ablation process are reported in Figure 1
216 with the main photoemission features from Fe, Si, Mg and O highlighted. Detailed scans of the
217 Fe $2p$, Si $2p$ and Mg $2s$ energy regions before and after the ablation process are reported in Figure
218 2. The binding energy scale is corrected by setting the binding energy (E_B) position of the C $1s$
219 peak from adventitious carbon to $E_B = 284.8$ eV as in Dukes et al.²⁰. In Figure 2, the photoemission
220 spectra are normalized to the intensity of the Si $2p$ lineshape.

221 After the ablation process, a drop in the Mg intensity is clearly visible in Figure 1 (consider,
222 for instance, the signal from the Mg $1s$ and Mg Auger lines). Consistently, the relative intensity of
223 the Mg $2s$ signal decreases in Figure 2b concomitant to an increase of the relative intensity of the
224 Fe signal (Figure 2a). The binding energy positions of the photoemission features of Figure 2 are
225 in agreement with the literature²⁰ and we can exclude the presence of metallic Mg, Si and Fe on
226 the sample surface since that XPS spectra show no signals related to these phases. The Fe $2p$
227 lineshape (Figure 2a) changes upon ablation, with a shift of the main photoemission peaks towards
228 higher binding energy and the disappearance of the charge transfer screening structure
229 (characterized by a satellite peak at about 715.5 eV⁴⁸⁻⁵⁰ related to the presence of Fe²⁺ ions in the
230 olivine matrix). This observation, together with the absence of any characteristic satellite structure,
231 points towards a partial oxidation of Fe²⁺ into Fe³⁺ upon ablation, consistent with the formation of
232 magnetite on the sample surface^{50,51}.

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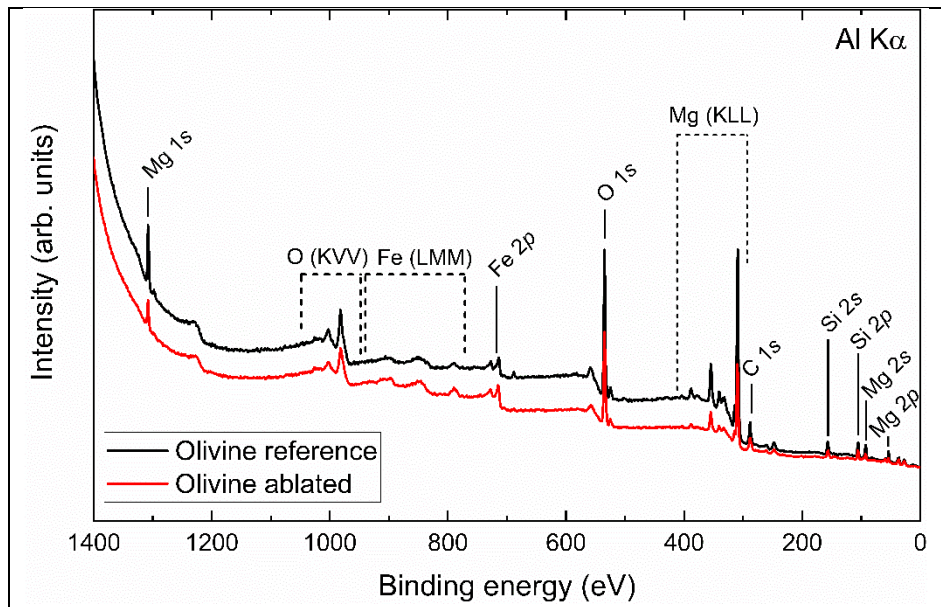


Figure 1. XPS wide energy scans acquired before (black) and after (red) the ablation process of the olivine single crystal. Dashed lines highlight the features related to Auger electron emission.

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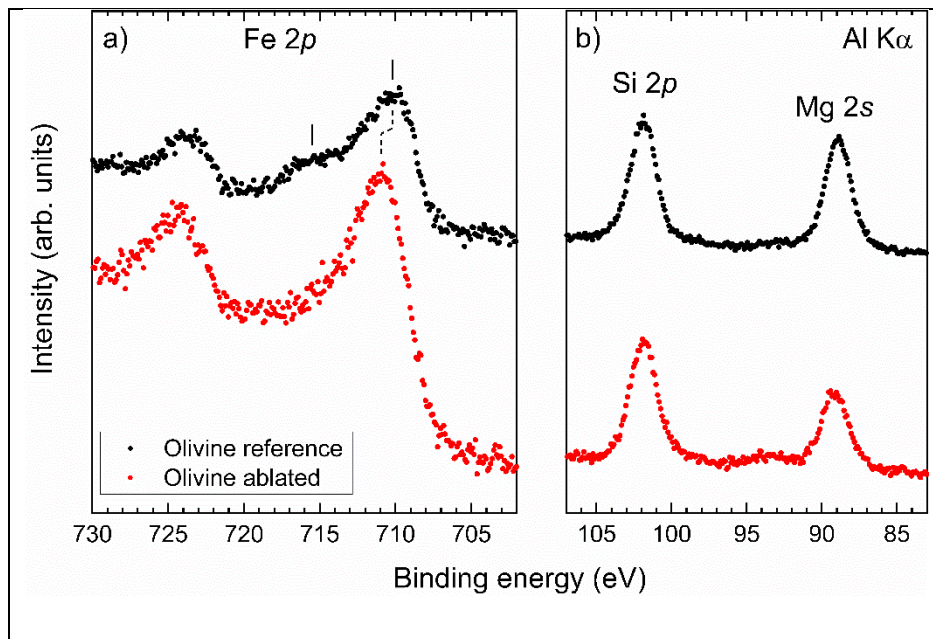


Figure 2. XPS spectra acquired on the a) Fe 2*p* and b) Si 2*p* and Mg 2*s* energy regions before (black) and after (red) the ablation process. Spectra have been normalized to the Si 2*p* photoemission intensity. Vertical lines in panel a) mark the energy position of the Fe 2*p*_{3/2} photoemission peak and its satellite.

235

236

237

238 **Optical absorption spectroscopy**

239 Optical absorption measurements on the resulting colloidal suspension composed by the
240 material ejected from the olivine surface into the water show a main band around 218 nm. This
241 feature is very similar to the “B₂ band” reported in several studies on silica glass^{52,53} and ascribed
242 to oxygen vacancies. This band is also a common feature in colloidal suspensions of synthetic
243 forsterite, Mg/Al-hydroxides and chrysotile. Notably, the band peak is subjected to a slight shift
244 due to the presence of different cationic species in addition or in substitution to silicon. In
245 particular, the main absorption signal shows a gradual shift from 217 nm (silica mesostructured
246 nanoparticles suspension signal) to about 218-219 nm when colloidal suspensions of synthetic
247 forsterite, ablated olivine nanoparticles and chrysotile are measured. While the shift is larger when
248 Mg- (222 nm) and Al- hydroxides (224 nm) are analysed (see Figure S2). Therefore, this behavior
249 suggests a dependance of the signal on both the coordination and valence of the cation species.
250 Moreover, the colloidal solution of olivine ejecta shows an additional and characteristic absorption
251 band around 270 nm, which is sensitive to pH. Indeed, it shifts progressively to 300 nm as the pH
252 is reduced by the addition of a diluted solution of sulfuric acid (Figure S3). This signal could be

253 due to the presence of SiO₂ nanoparticles together with ferric and ferrous oxides. These latter are
254 dissolved by sulfuric acid liberating ferric and ferrous ions which, at acidic pH, give rise to
255 absorption bands centered at 300 nm (see ref ⁵⁴ SiO₂ nanoparticles; see refs ^{55,56} for magnetite
256 nanoparticles and ⁵⁷).

257

258 **High-resolution transmission electron microscopy**

259 After the LP-PLA and optical spectroscopy analyses, the colloidal suspension was (i) drop-
260 casted (2μl) onto a C-coated Cu grids and (ii) subjected to a drying process and re-dispersion in
261 ethanol for the application of a constant electric field to promote electro-deposition of suspended
262 nanoparticles onto C-coated Cu grids serving as anode and cathode. These grids were then
263 analyzed by means of HR-TEM/EDS.

264 **Drop-cast grid**

265 Detailed HR-TEM imaging analyses on the drop-cast of the ejecta revealed the structural
266 complexity of the ablated material. The size distribution varies from large globular aggregates of
267 hundreds of nanometers in diameters (~600 nm) to small nanoparticles of 4-10 nm in diameter
268 (Figure 3 a-f) as also confirmed by the DLS analyses (see Figure S4). Moreover, large crystalline
269 fragments of hundreds of nanometers are also present (Figure3 g-i).

270

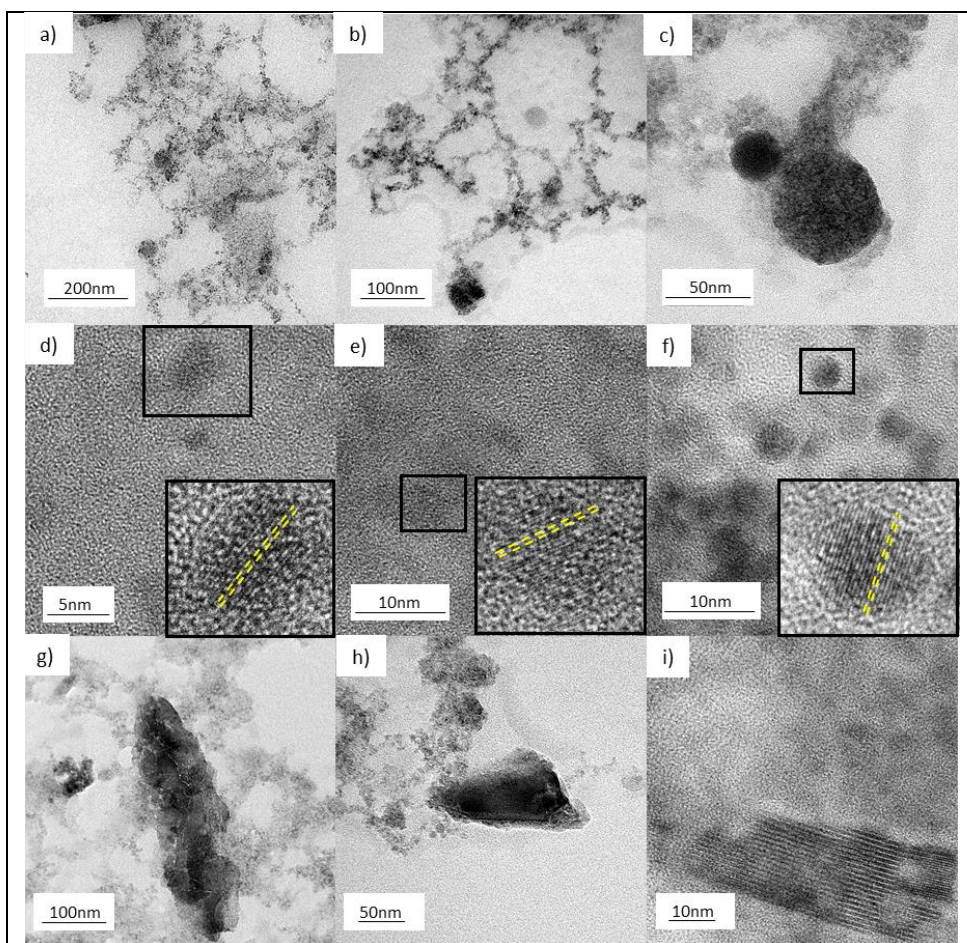


Figure 3. Bright field images of the drop-cast colloidal suspension: (a,b,c) Globular nanoparticle aggregates; (d,e,f) high-resolution images of the aggregates in (a) showing the individual nanoparticles of about 4 to 10 nm in diameter and their lattice fringes as evident in the insets; (g,h,i) crystalline fragments.

271
 272 High-resolution bright field images were collected in the region nearby the globular
 273 aggregates (as those reported in Figure 3 a-c) revealing the presence of spherical particles at the
 274 nanoscale (see Figure 3 d-f). The nanoparticles show evidence of lattice fringes pointing to their
 275 crystalline nature. Because of the low concentration of nanoparticles within the selected area

276 aperture, their very small dimensions, and the high background due to the carbon membrane and
277 amorphous material in general, diffraction rings were discontinuous and diffuse, and could not be
278 profitably exploited. The interplanar distances of the crystalline nanoparticles were therefore
279 obtained by the lattice fringe spacing in high-resolution images, which gave a systematic value
280 around $2.1 \pm 0.1 \text{ \AA}$ (value averaged on 12 measurements). Unfortunately, this datum alone was not
281 sufficient to univocally identify the nanoparticles, since most candidate phases have at least one
282 family of crystallographic planes with distances in this range. EDS spectra collected on the
283 nanoparticle aggregates show the presence of Fe, Mg, Si and O with sometimes a slight enrichment
284 in Fe with respect to Mg, indicating a compositional change of the nano-ablated product with
285 respect to the parent olivine (see Figure S5). On this basis, and considering the measured average
286 d-spacing of $2.1 \pm 0.1 \text{ \AA}$, periclase (most intense reflection at 2.11 \AA) and wustite (most intense
287 reflection at 2.15 \AA) are among the possible phases forming the ejecta material.

288 On the other hand, the average measurable interlayer spacings of the large fragment
289 displayed in Figure 3i are $10.2(3) \text{ \AA}$ and $4.8(2) \text{ \AA}$, which are consistent with (010) and (100) planes
290 of olivine, respectively (Figure 4a). This same fragment also shows some clear effects of the
291 ablation process such as local lack of periodicity and misorientation of the lattice planes (see Figure
292 4a). These features, along with the resulting EDS spectrum (Figure 4b), are consistent with
293 fragments of ablated olivine.

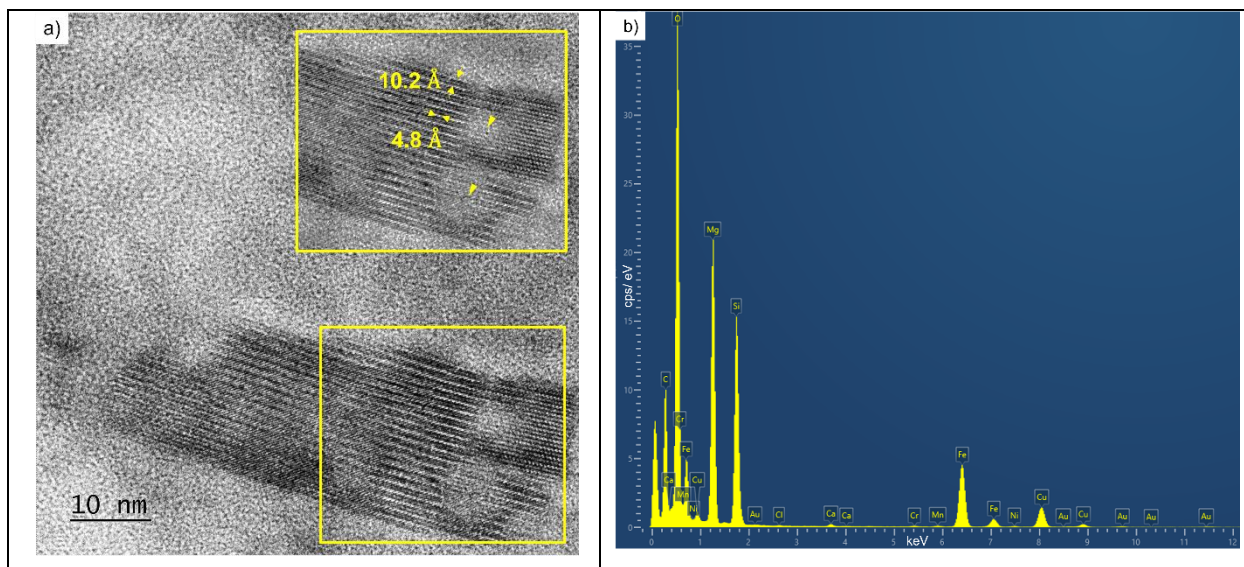


Figure 4. a) Large crystalline fragment found in the ablated product. It shows d-spacings of 10.2(3) and 4.8(2) Å consistent with the (010) and (100) planes of olivine, view along [001]. Effects of the ablation process are also evident as planes misorientation and lack of periodicity (yellow arrows in the top right inset). **b)** EDS spectrum.

294

295 Electro-deposition grids

296 Electro-deposition was performed in order to exploit the electrostatic properties of the
 297 suspended nanoparticles with the aim to promote their separation and, consequently, to allow for
 298 a better chemical characterization. The analysis of the ζ -potential of the ablated nanoparticles re-
 299 suspended in ethanol performed by dynamic light scattering methods showed a clear population
 300 of charged particles centered at -30 mV (see Figure S6). Despite this evidence, after electro-
 301 deposition, nanoparticles were revealed in both the cathode and anode C- coated Cu grids (Figure
 302 5). Moreover, the electro-deposition promoted the formation of a homogeneous layer of
 303 nanoparticles with sometimes sparse branched big aggregates (see Figure S7).

304 On the cathode, nanoparticles are found with an average diameter size of 4 ± 1 nm, while
305 on the anode the size of the nanoparticle was 9 ± 3 nm (Figure 5a and c). The nanoparticles on
306 both the electrodes give neater signals with respect to those of the drop-cast allowing for a
307 diffraction analysis.

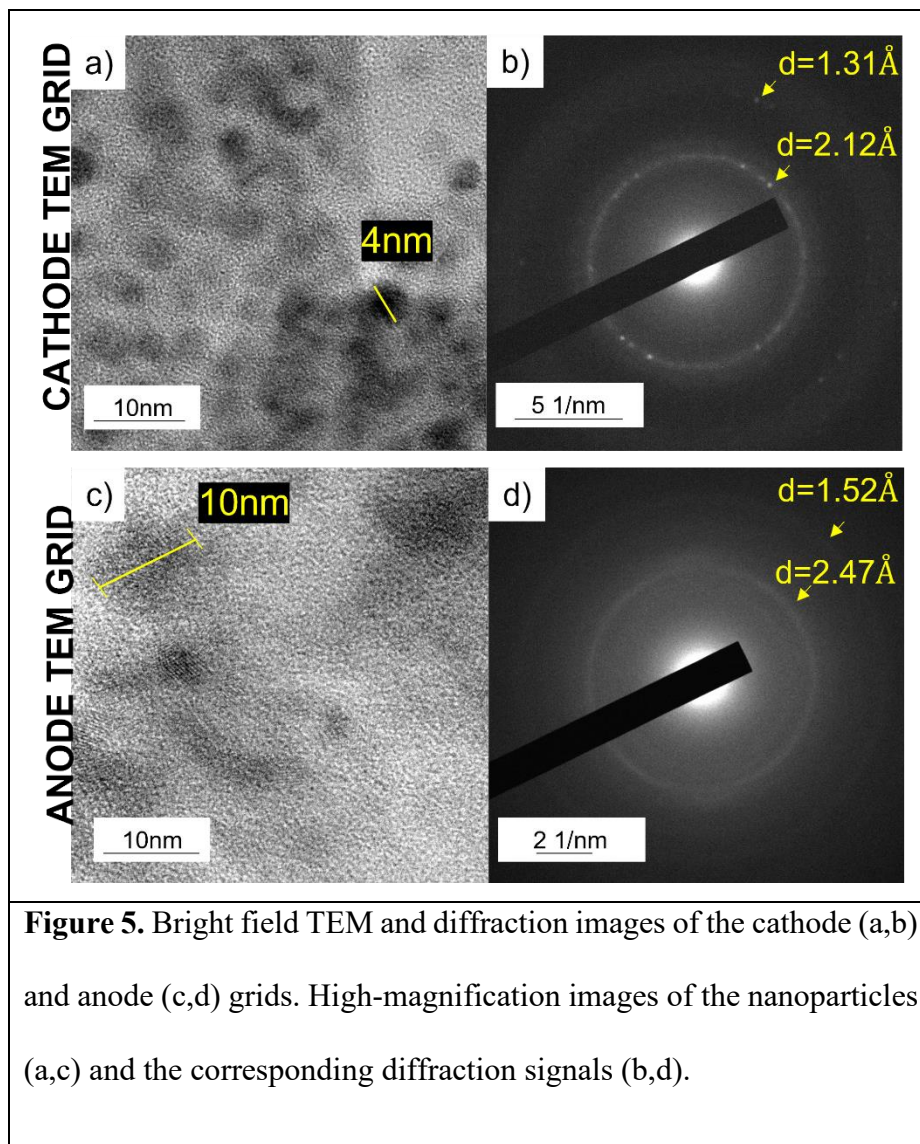
308 The SAED patterns acquired on the nanoparticle aggregates reported in Figure 5b and d
309 show the coexistence of an amorphous fraction (diffuse haloes) together with a crystalline one
310 (weak diffraction rings and spotted rings, see yellow arrows). Interlayer spacings of the crystalline
311 nanoparticles have been determined from (i) HR lattice fringe images and (ii) SAED patterns.
312 Among all the measured values four main d-spacings have been measured: 1.32 ± 0.01 Å (average
313 of 2 measurements), 1.60 ± 0.01 Å (average over 3 measurements), 2.13 ± 0.06 Å (average of 44
314 measurements) and 2.52 ± 0.05 Å (average of 17 measurements). Moreover, in high-resolution
315 bright-field images, each individual crystalline nanoparticle shows only one lattice fringe
316 corresponding to one of the four aforementioned d-spacings, while in the collected SAED patterns
317 up to two simultaneous reflections among the four reported above can be observed (see yellow
318 arrows in Figure 5b and d). The measured interlayer spacings are consistent with the most intense
319 X-ray reflections of wustite, periclase, metallic Mg, magnetite, hematite, magnesioferrite and
320 maghemite (see highlighted values in Table S2). The diffuse haloes are located close to the
321 diffraction rings and have average d-spacings of 2.1 ± 0.1 Å, 1.2 ± 0.2 Å (Figure 5b) and 2.5 ± 0.1
322 Å, 1.56 ± 0.2 Å (Figure 5d), similar to those measured for the crystalline fraction^{58,59}.

323 Therefore, in order to narrow the range of the possible phases, we collected EDS spectra
324 on both the electrodes in areas with a sufficient concentration of nanoparticles. The chemical
325 analyses revealed the presence of only Mg, Si and O, whereas Fe was below the detection limit

326 (see Figure S8). The higher amount of Mg with respect to Fe together with the measured d-spacings
327 point out to a possible presence of periclase nanoparticles as dominant phase in the ejecta material.

328 Furthermore, it is worth noting that, despite showing the same morphology and chemical
329 composition, the nanoparticles were found with smaller size on the cathode, while with larger size
330 on the anode. This points to a dependence of the nanoparticle charge on its size due to their high
331 surface-to-volume ratio ^{40,60,61}.

332



333

334

335 DISCUSSION

336 Our multi methodological approach, combining the analysis of both the ablated material
337 and the target, showed the occurrence of clear structural and chemical changes as a consequence
338 of the simulation of space weathering in water environment. Specifically, the ablated material
339 (ejecta) is made up by (i) a fraction of crystalline nanoparticles (4-10 nm in diameter) with
340 diffraction signals mostly compatible with the presence of metallic Mg and/or periclase as the
341 dominant phase with minor Fe-oxides; (ii) amorphous material (as evident from the diffuse halo
342 in the diffraction images) with a composition close to the crystalline fraction and (iii) shocked
343 crystalline olivine fragments of size of the order of 500-600 nm (see Figure 3 g-I and Figure 4).

344 On the other hand, the target is depleted in Mg and enriched in Fe, showing a partial
345 oxidation of Fe^{2+} into Fe^{3+} , in agreement with the formation of magnetite domains on the sample
346 surface.

347 The evidence of partial oxidation of Fe^{2+} into Fe^{3+} on the target surface is in net contrast
348 with the results of similar ablation experiments on olivine samples performed in vacuum^{17,18,22},
349 where the authors report the occurrence of metallic Fe-nanoparticles. The presence of metallic iron
350 particles together with the Mg removal from the target is explained by the thermal reduction
351 mechanism in absence of a subsequent re-oxidation (see ref²² for more details). Chemical
352 reduction is a common process also in water, promoted by hydrated electrons forming during
353 plasma ignition which are potentially able to reduce both Fe and Mg into their metallic state⁴⁰.
354 These species might survive in water (or even only in air) as nanoparticles only if immediately
355 capped by an inert layer. This layer has been demonstrated to be represented by silica in laser
356 ablation experiments performed on Si targets immersed in metallic precursor solutions (Ag^+ ,

357 Au³⁺). These studies have shown that metastable species such as H₂SiO₂ can be formed as the
358 result of the interaction of Si-rich targets and water, exhibiting the capability of reducing noble
359 metal ions into metallic nanoparticles while oxidizing into silica forming a capping layer around
360 them^{57,62-66}. In our case, this process might be active in a region relatively far from the solid-liquid
361 interface, while re-oxidation is promoted by the presence of radical species in the water
362 environment in a region confined at the solid-liquid interface⁶⁷. This might account for the fact
363 that we have nanoparticles compatible with metallic Mg. However, by the analysis of the lattice
364 spacing of the produced nanoparticles, we can confidently rule out the presence of metallic Fe.
365 This observation can be explained by an apparent tendency of Fe-ions to remain bonded to the
366 target, in contrast to the tendency of Mg-ions to detach from the target, as demonstrated by the
367 increase in the Fe-related signal and the simultaneous decrease in the Mg-related signal in XPS
368 spectra (Figure 2).

369 Since the raster scanning of the target during the ablation process was performed by
370 moving the stage manually, it is likely that the newly formed magnetite on the sample surface was
371 also subjected to a subsequent ablation with its further transport into the colloidal suspension.
372 Indeed, the measured spacings of the crystalline nanoparticles are also consistent with the presence
373 of magnetite, in addition to the other abovementioned possible Mg and Fe phases (see TEM results
374 session). However, the presence of Fe has been detected only in the drop-cast material, while it
375 was below the detection limit on both cathode and anode grids (Figure S8). Therefore, the d-
376 spacings measured on both cathode and anode TEM grids, even if compatible with the presence of
377 magnetite and other Fe compounds (see Table S2), are likely due to Mg compounds (i.e., silica-
378 capped metallic Mg and periclase). This evidence indicates that Fe-bearing nanoparticles do not

379 acquire a substantial charge enabling their migration to the electrodes when re-dispersed in
380 ethanol.

381 The presence of iron oxides (e.g., magnetite) and therefore the absence of metallic iron is
382 consistent with an environment that experienced aqueous alteration during its history as occurred
383 for example on Mars or on carbonaceous chondrites parent body ⁶⁸⁻⁷¹. Moreover, in the outer space,
384 a large number of bodies is (or has been) characterized by the presence of volatile-rich elements
385 as well as hydrous minerals or water ices (e.g., Mars, Ceres, asteroids, comets) that affect the
386 chemistry and physics of space weathering processes ^{15,29,72}. In particular, magnetite is quite
387 ubiquitous on Earth and in planetary materials and it is often a marker for a variety of processes
388 as for water-rock interactions (e.g., serpentinization, see ref ⁷³ and references therein).

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391 CONCLUSIONS

392 The coupling of the nanosecond liquid-phase pulsed laser ablation with the use of olivine
393 single crystals as targets allows us to characterize both the shocked target and the ejecta material
394 (i.e., colloidal suspension).

395 We provide a methodological approach to investigate space weathering processes in a
396 water environment. We observed that the water environment triggers the partial oxidation of Fe²⁺
397 into Fe³⁺ in a region confined at the solid target-liquid interface, promoting the formation of
398 magnetite on the sample surface. On the other hand, Mg is easily removed, and the target results
399 Mg-depleted (Fe-enriched) after the ablation process.

400 In the ablated material, we find crystalline fragments close to an olivine composition (i.e.,
401 starting material) together with amorphous and crystalline nanoparticles. The crystalline

402 nanoparticles have diffraction signals compatible with the mainly occurrence of Mg- bearing
403 phases (i.e., Mg metallic and periclase) with minor Fe- phases: from silica-capped metallic Mg to
404 magnetite. The latter probably coming from the ablation of the newly oxidized iron at the solid-
405 liquid interface. The diffuse diffraction signals present in the diffraction images (Figure 5) together
406 with the EDS analyses are also consistent with the occurrence of the same phases, as reported
407 above for the crystalline nanoparticles, but in an amorphous status.

408 The identification of magnetite on the olivine surface together with the occurrence of iron
409 oxides and metallic Mg capped by a silica shell in the ablated material might represent evidence
410 of a space weathering processes associated with an aqueous environment on planetary materials
411 and asteroids. Moreover, it seems clear that iron has a weak tendency to escape the olivine surface
412 and that the water environment triggers its oxidation. This is expected to affect reflectance spectra
413 through the absence of features related to metallic iron nanoparticles and the lack of reddening
414 phenomena. Moreover, this has profound implications in several research fields from astrobiology
415 to planetary geology.

416 Even if a lot of work is still required to better identify the ablated species in the ejecta
417 material, we provide an unexplored procedure in the field of space weathering studies of
418 astronomical silicates. This study also contributes to the development of the new research area
419 focused on the study of space weathering and alteration processes in different environments with
420 respect to those of the airless silicate bodies in the inner solar system.

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425 ASSOCIATED CONTENT

426 **Supporting Information**

427 The Supporting Information is available free of charge at:

428 Chemical analyses (Table S1), Mössbauer spectra (Figure S1), Absorption spectra (Figure S2),
429 Absorption spectra with addition of known aliquots of sulfuric acid (Figure S3), Dynamic light
430 scattering measurements (Figure S4), TEM/EDS chemical analyses (Figure S5), Zeta potential
431 measurement (Figure S6), Low magnification TEM images (Figure S7), d-spacing of reference
432 chemical species (Table S2), EDS spectra (Figure S8) (PDF)

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453 **Notes**

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